
Characterisation of mesopores - Positronium lifetime measurement as a porosimetry technique

S. Thränert, D. Enke, R. Krause-Rehberg



Characterising mesopores

■ Principles of PALS

- Lifetime Measurement
- Positronium

■ Porous glass - CPG

- Synthesis
- Properties

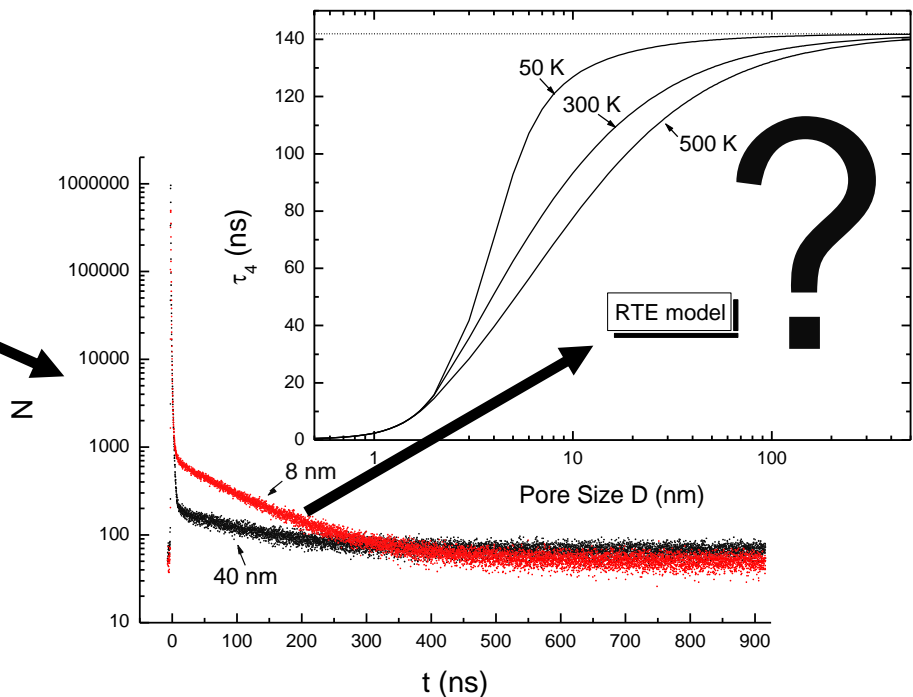
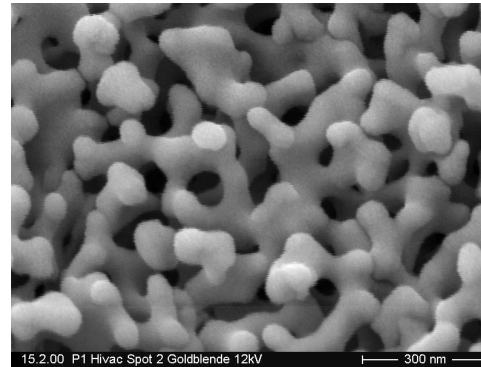
■ Models - the state of the art

- Tao Eldrup
- Tokyo
- RTE

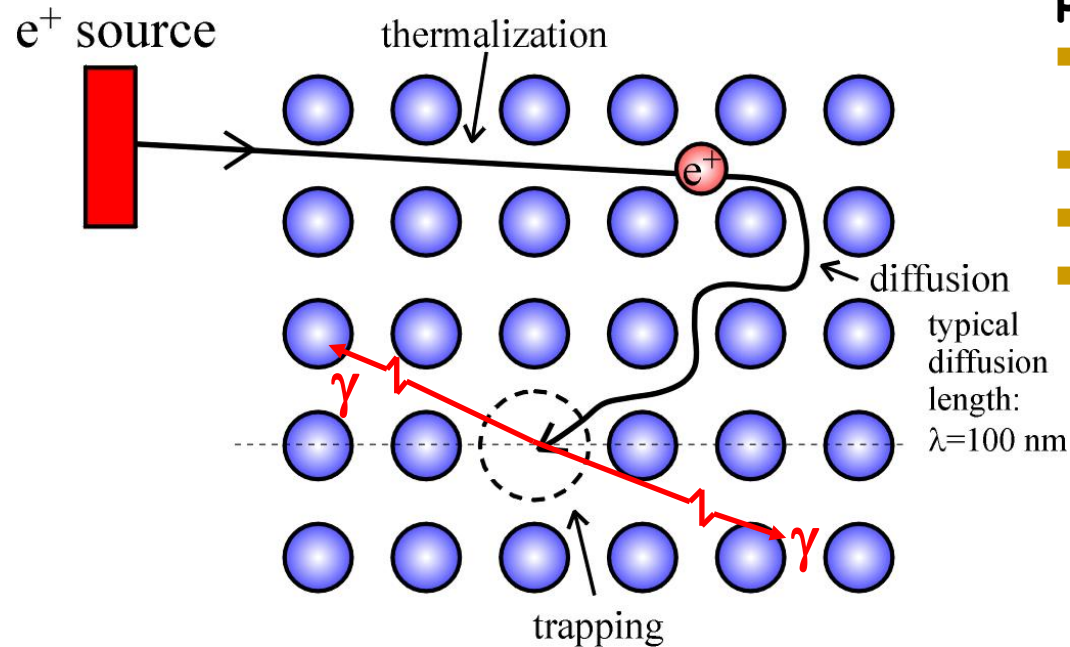
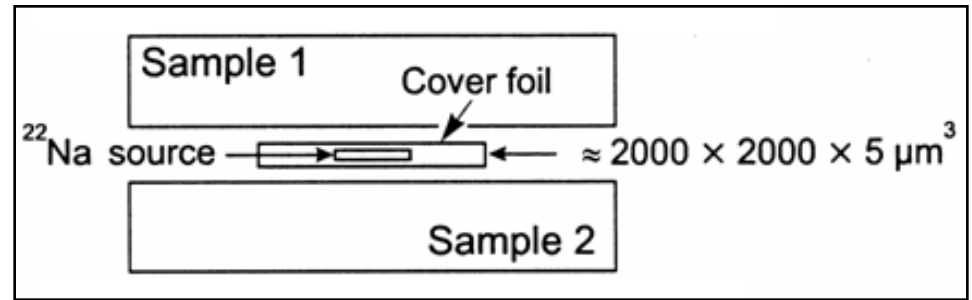
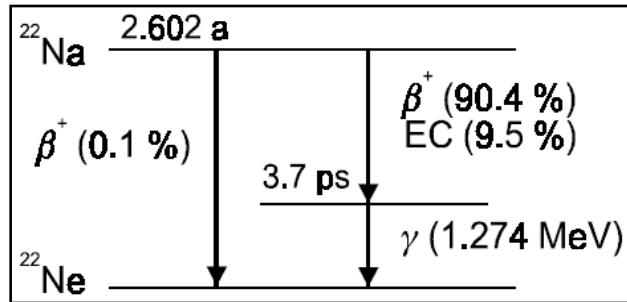
■ Experimental results

- Relation to RTE

■ Summary



Principles of PALS



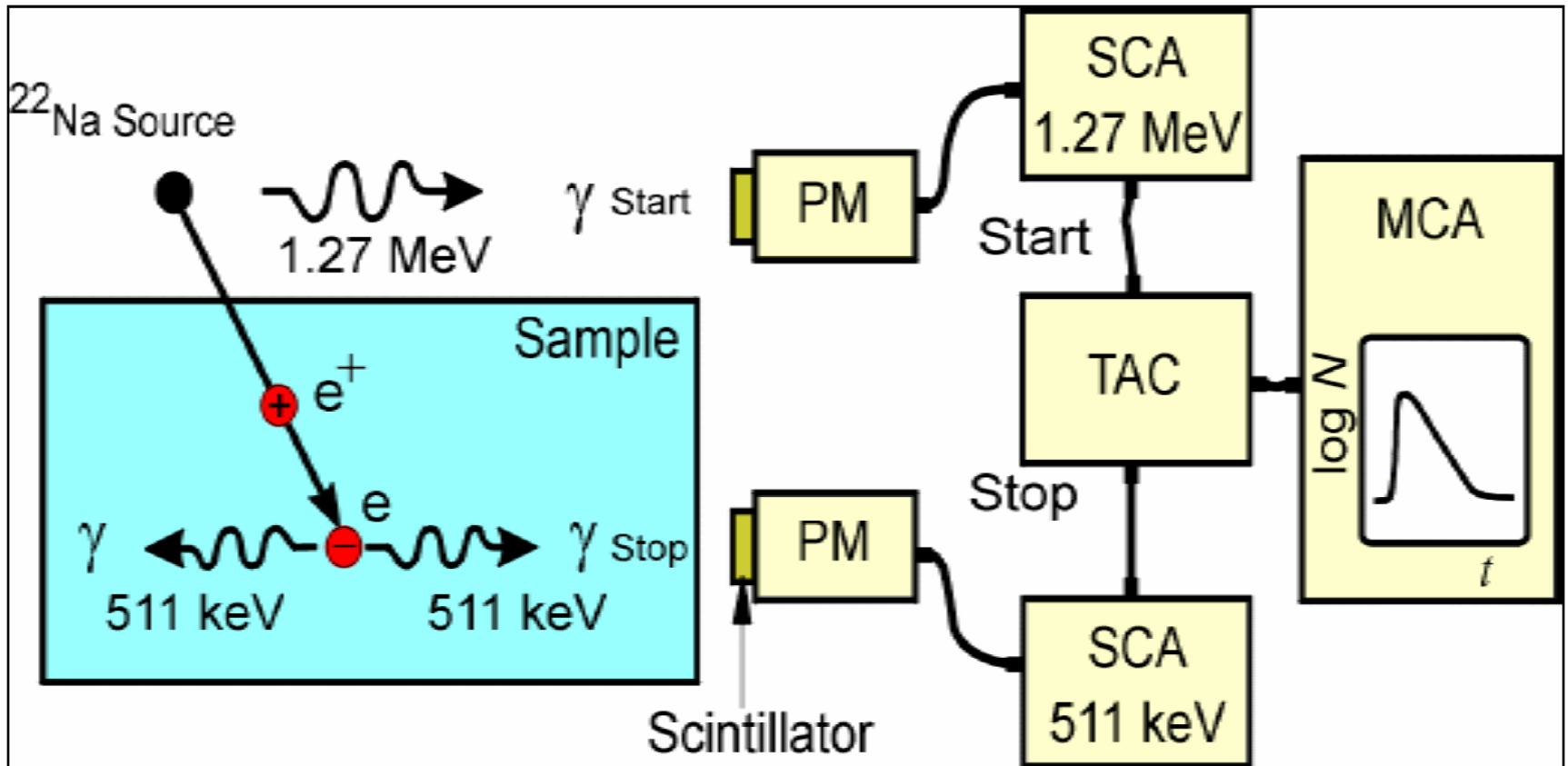
positrons:

- thermalize (reach thermal energies)
- diffuse
- being trapped
- and annihilate

When trapped in vacancies:

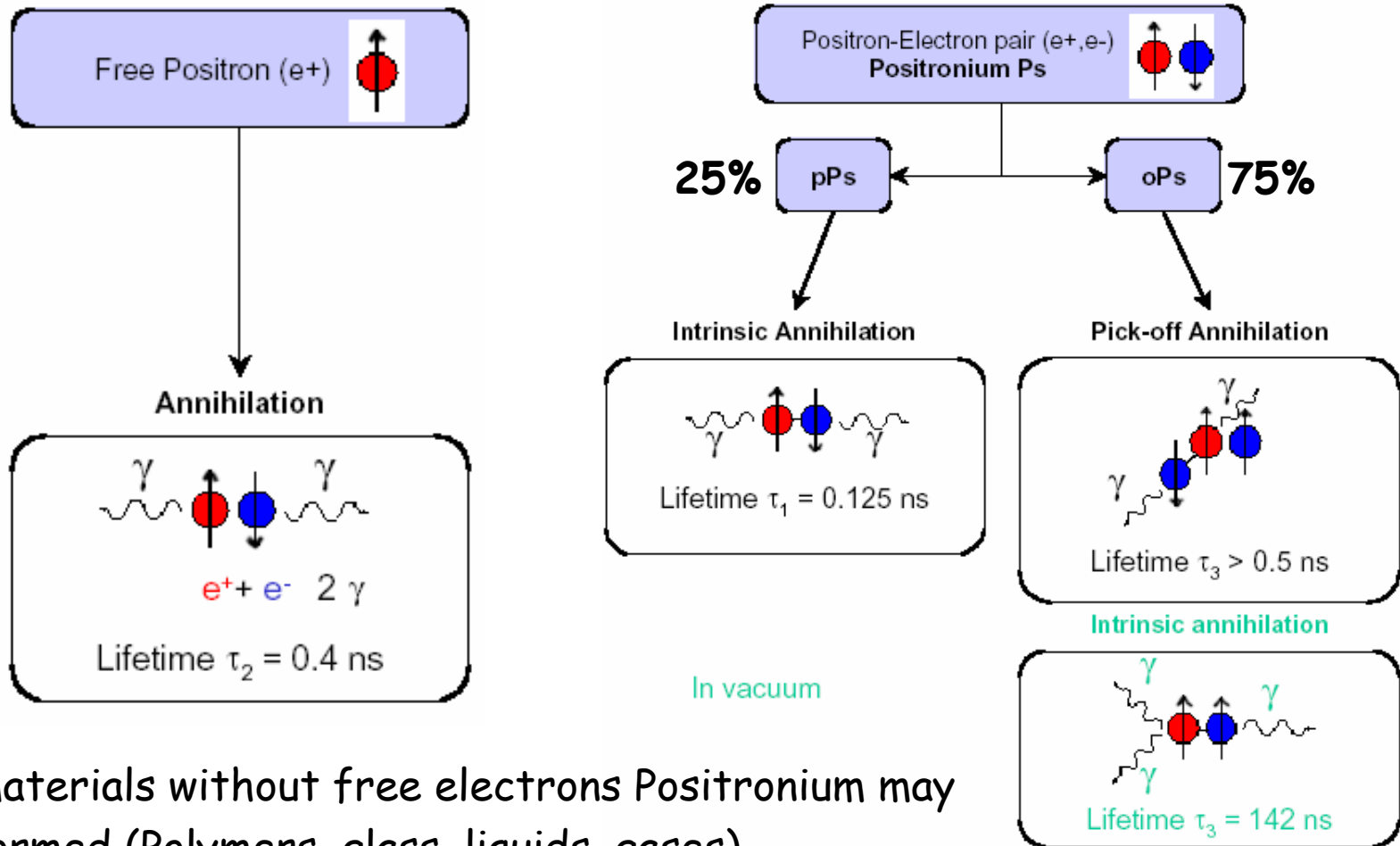
- Lifetime increases due to smaller electron density in open volume

Principles of PALS



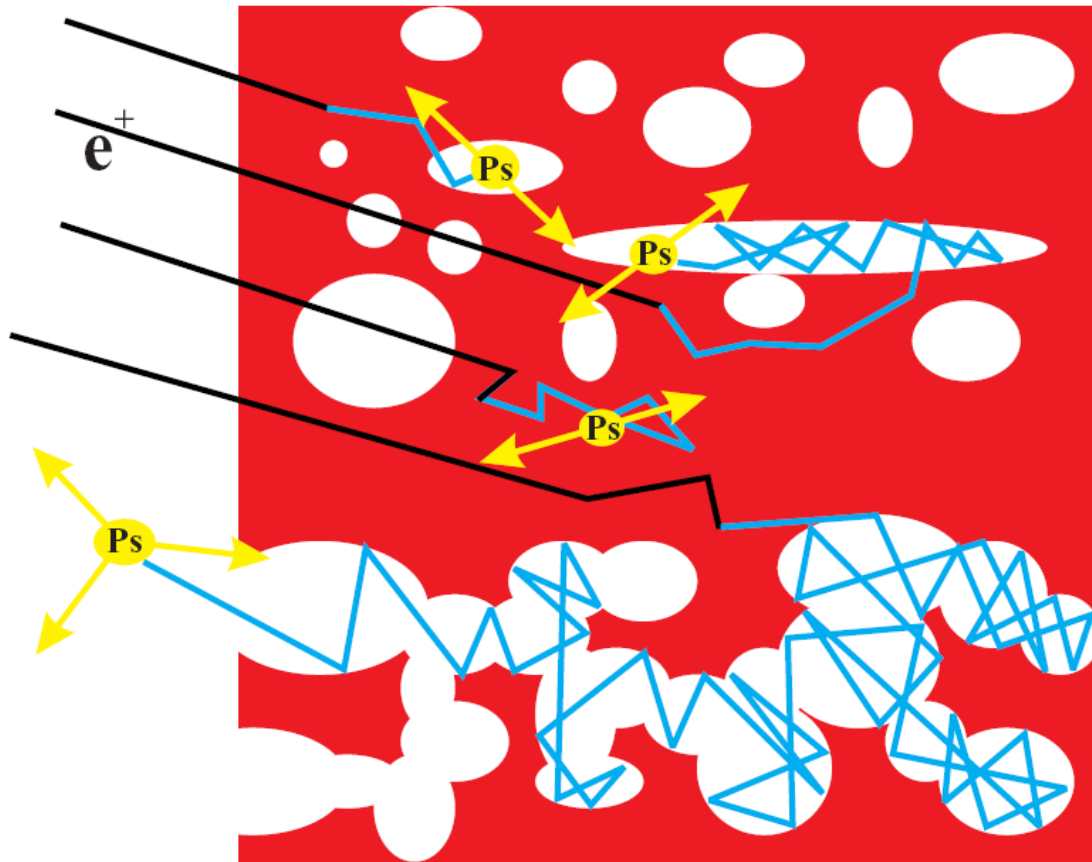
- Positron lifetime: time between $1,27\text{ MeV}$ and $0,511\text{ MeV}$ quanta

Principles of PALS: ortho-Positronium



- In materials without free electrons Positronium may be formed (Polymers, glass, liquids, gases).

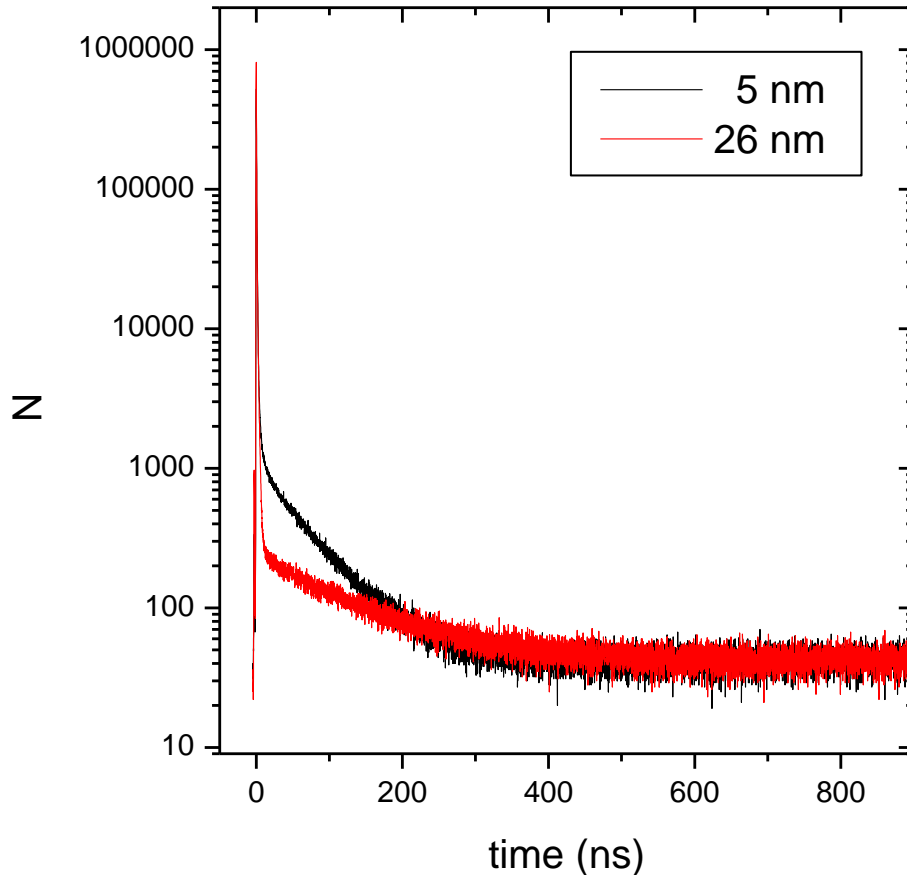
Principles of PALS: pick-off annihilation



pick-off annihilation:

- o-Ps is converted to p-Ps by capturing an electron with anti-parallel spin
- happens during collisions at walls of pore
- lifetime decreases rapidly
- lifetime is function of pore size 0.5 ns ... 142 ns
- lifetime can be extracted from spectra

Principles of PALS: typical spectrum

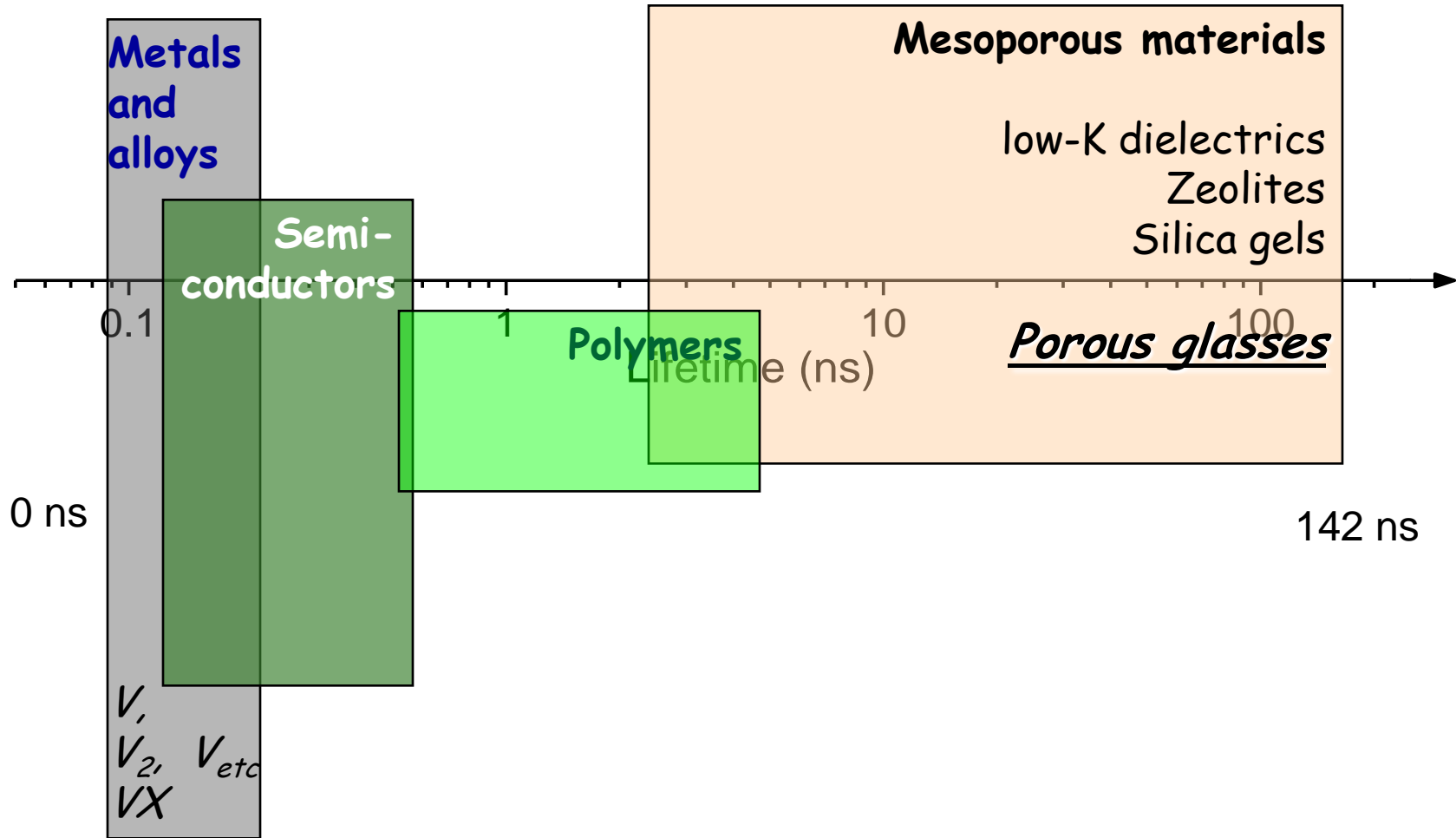


typical lifetime spectrum
for porous glass:

- 4 exponential decay components
- p-Ps \rightarrow 0.125 ns
- free positrons \sim 0.5 ns
- o-Ps in amorphous region of glass \sim 1.5 ns
- o-Ps in pores

$$N(t) = \sum_{i=1}^{k+1} \frac{I_i}{\tau_i} \exp\left(-\frac{t}{\tau_i}\right)$$

Principles of PALS: typical lifetimes



Characterising mesopores

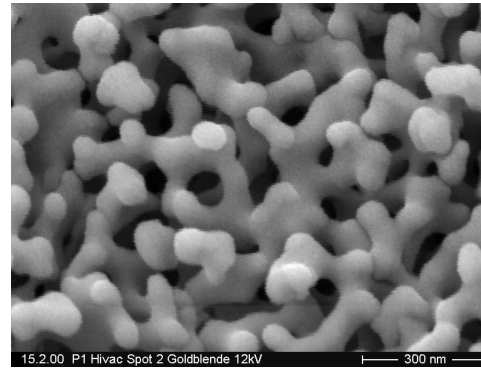
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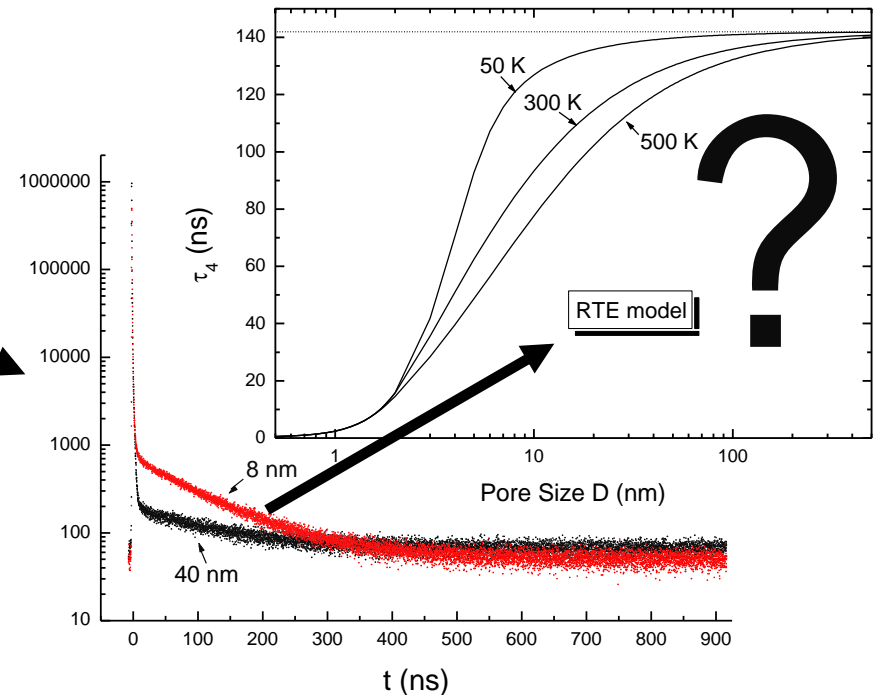
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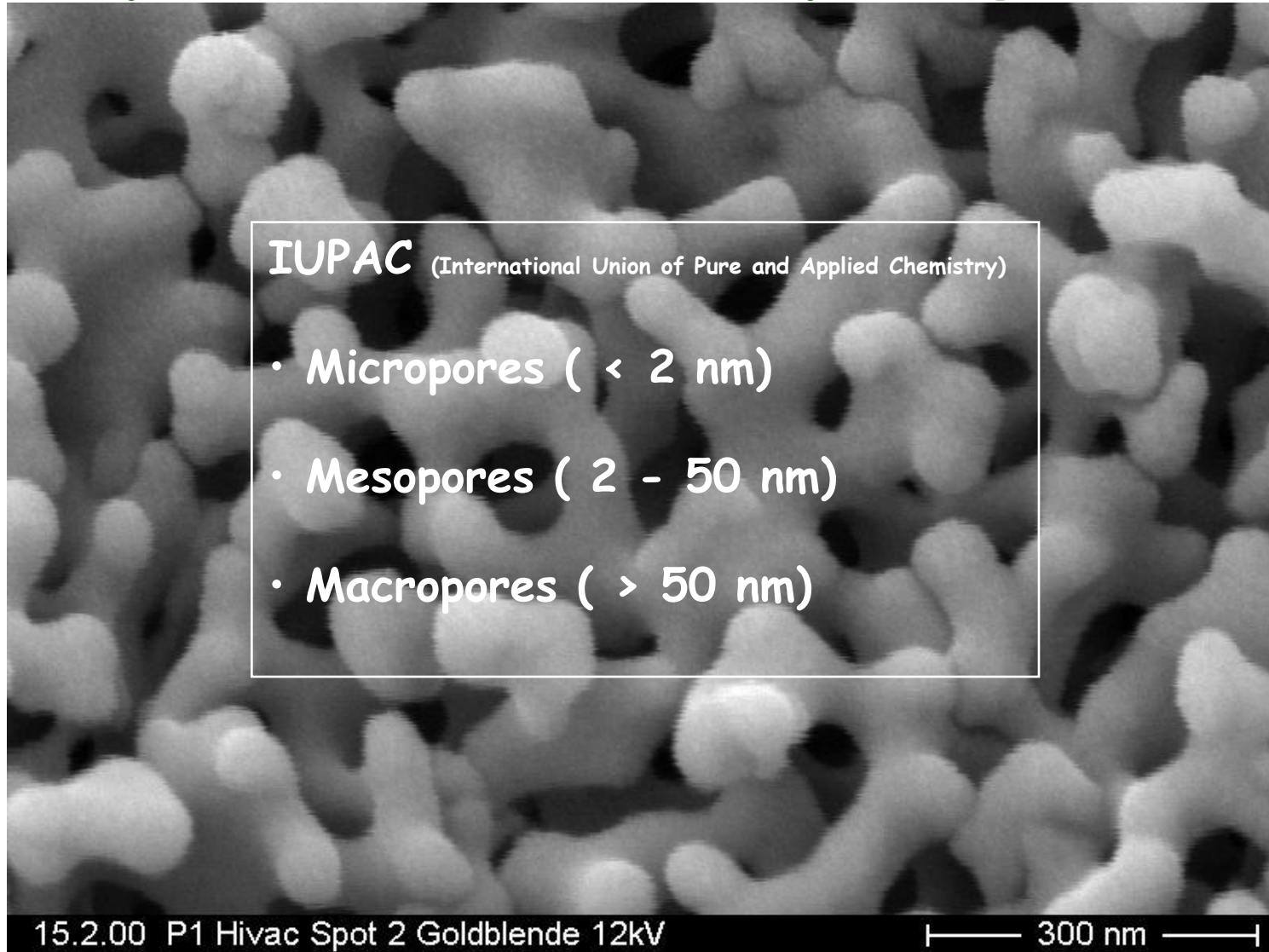
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Mesopores - Controlled pore glasses



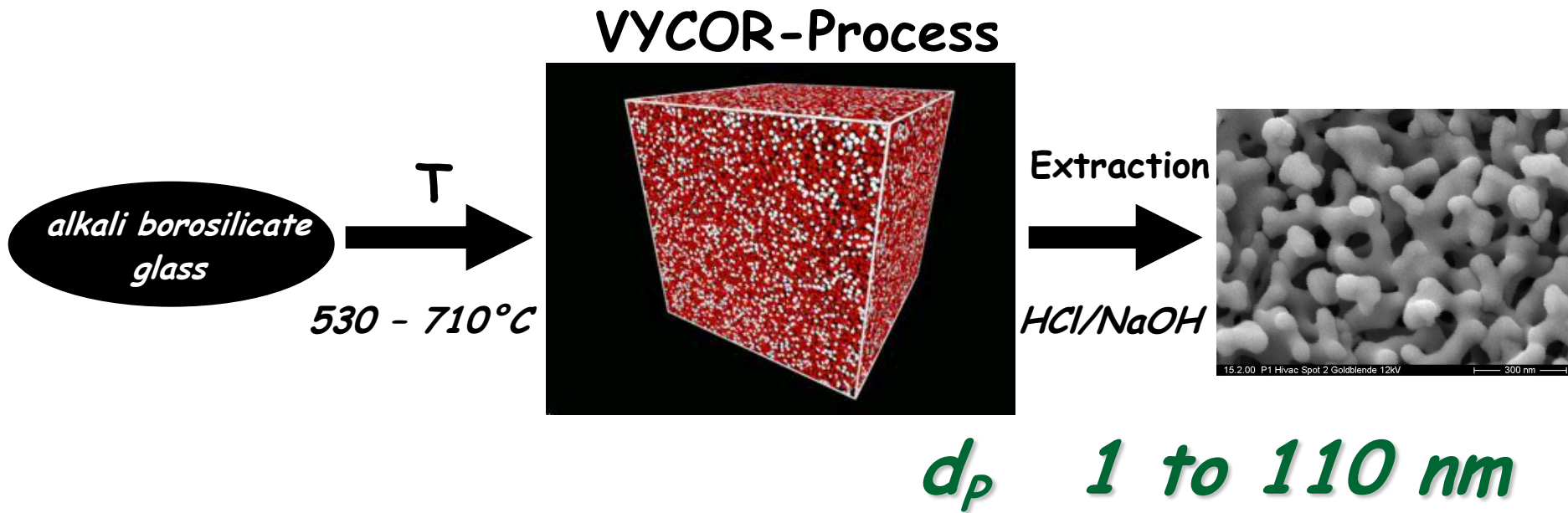
IUPAC (International Union of Pure and Applied Chemistry)

- Micropores (< 2 nm)
- Mesopores (2 - 50 nm)
- Macropores (> 50 nm)

15.2.00 P1 Hivac Spot 2 Goldblende 12kV

300 nm

Controlled pore glasses - CPG



d_p 1 to 110 nm

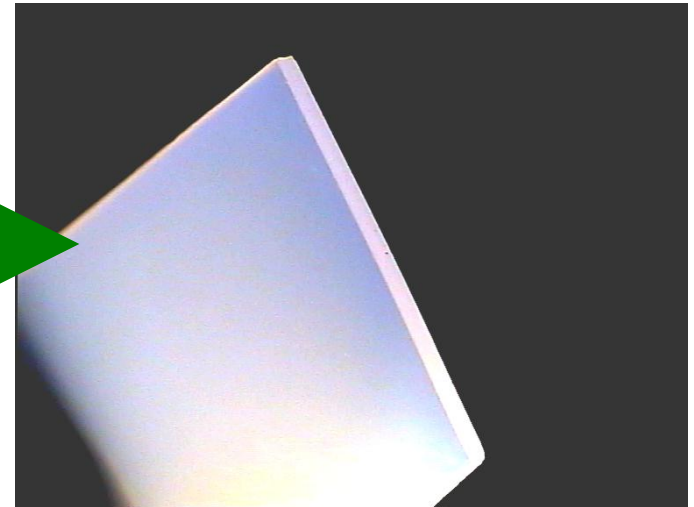
- spinodal phase separation
- decomposition is initiated by heat treatment
- alkali rich borate phase \leftrightarrow pure silica
- alkali phase soluble in acid \rightarrow silica network
- pore size depends on basic material
- shape depends on duration and T of heat treatment

Controlled pore glasses - CPG



porous microspheres:

- 100 μm

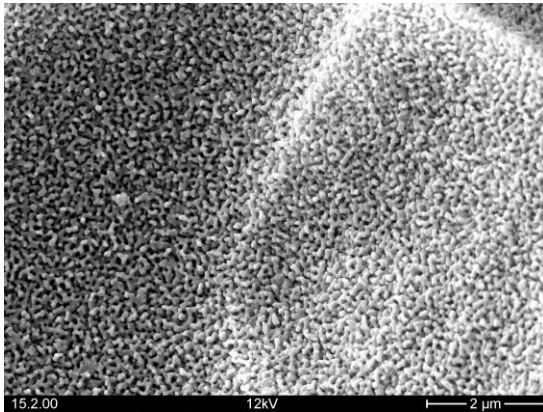


porous membranes:

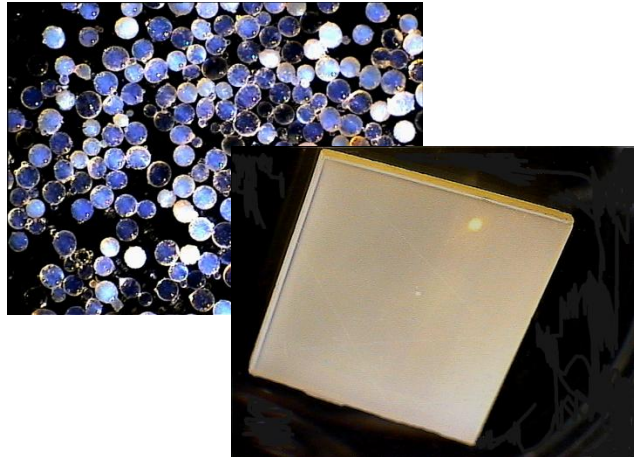
- 20 x 20 x 0.2 mm

DE-Patent 19848377 A1

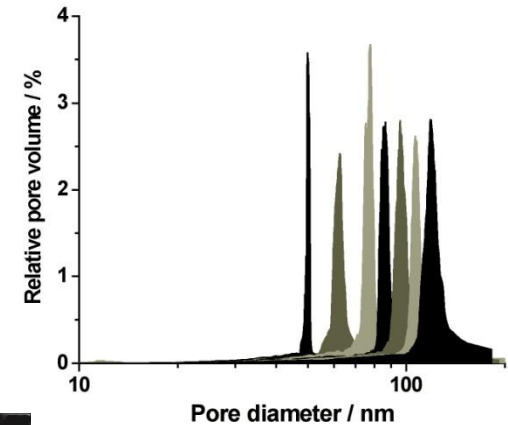
Controlled pore glasses - CPG



- different geometries possible



- homogenous microstructure
- small pore size distribution



- pore size arbitrary

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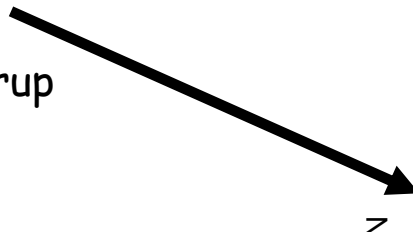
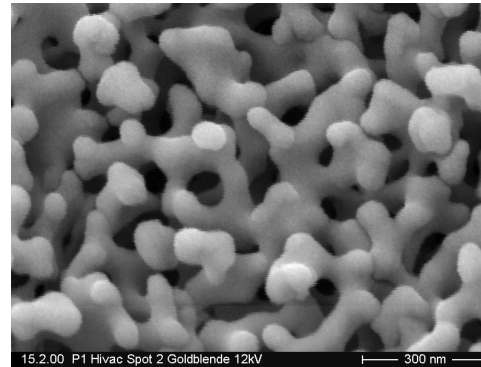
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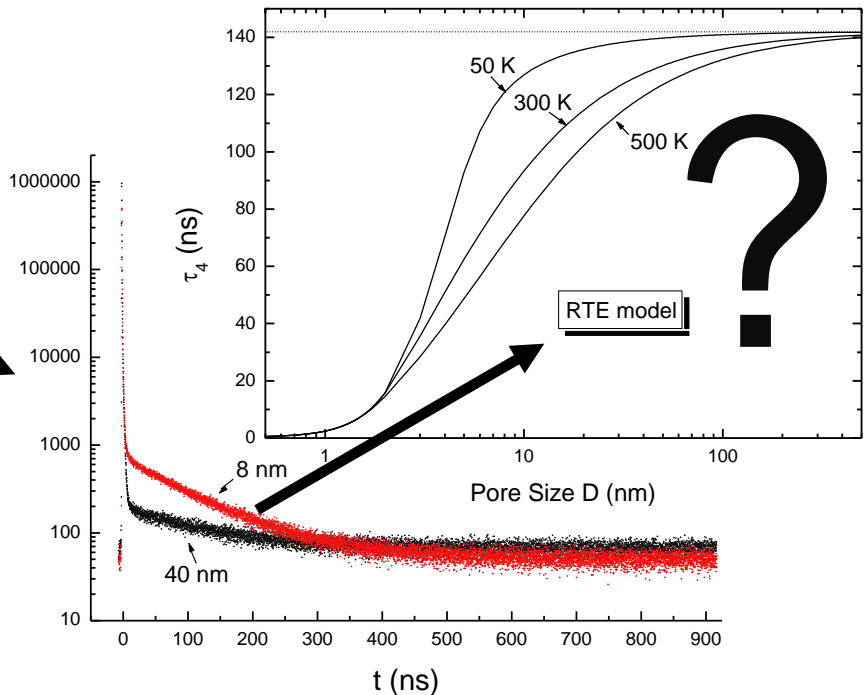
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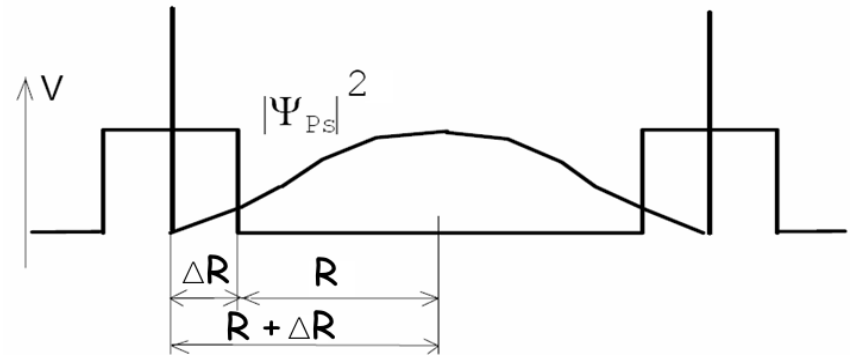
The TE model

- Annihilation rate: $\frac{1}{\tau_{o-Ps}} = \lambda_{o-Ps}$

$$= \lambda_{2\gamma} + \lambda_{3\gamma}$$

$$= \lambda_{2\gamma}^0(P) + \lambda_{3\gamma}^0(1-P) \cong \lambda_{2\gamma}^0(P)$$

$$\lambda_{2\gamma}^0 = \frac{\lambda_S + 3\lambda_T}{4} = \lambda_A \approx 2ns^{-1}$$

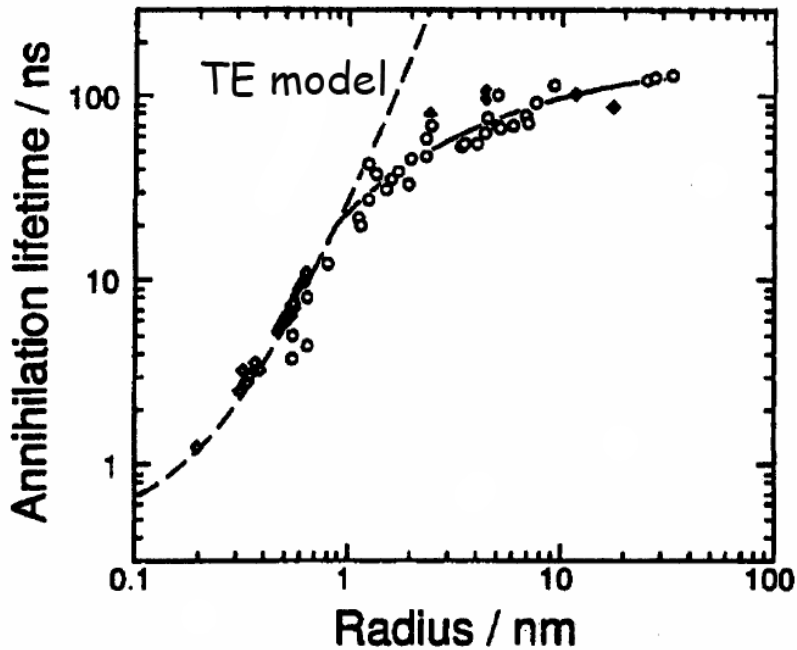


- Pore size < 1 nm → $\lambda_{3\gamma}$ neglected, only pick off annihilation

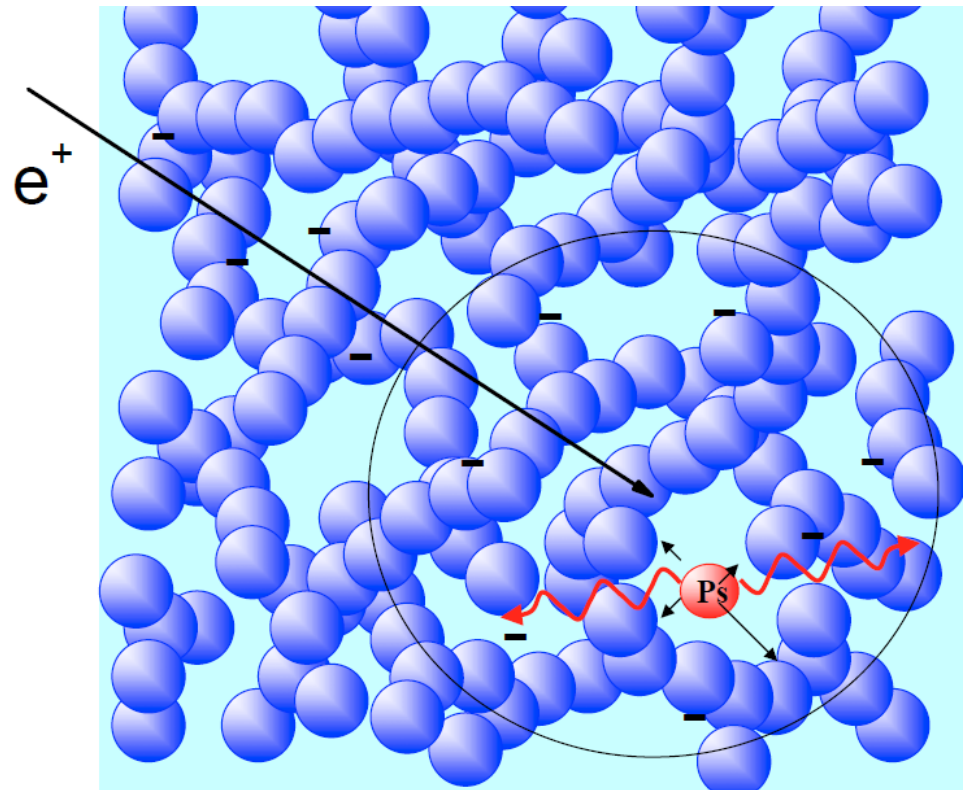
$$\lambda_{TE}(R) = \lambda_A \left[1 - \frac{R}{R + \Delta R} + \frac{1}{2\pi} \sin\left(\frac{2\pi R}{R + \Delta R}\right) \right]$$

- $\Delta R = 0.166$ nm determined by Eldrup and Jean
- Pore size > 1 nm → $\lambda_{3\gamma}$ can not be neglected, temperature dependence of o-Ps lifetime (excited states)

The TE model

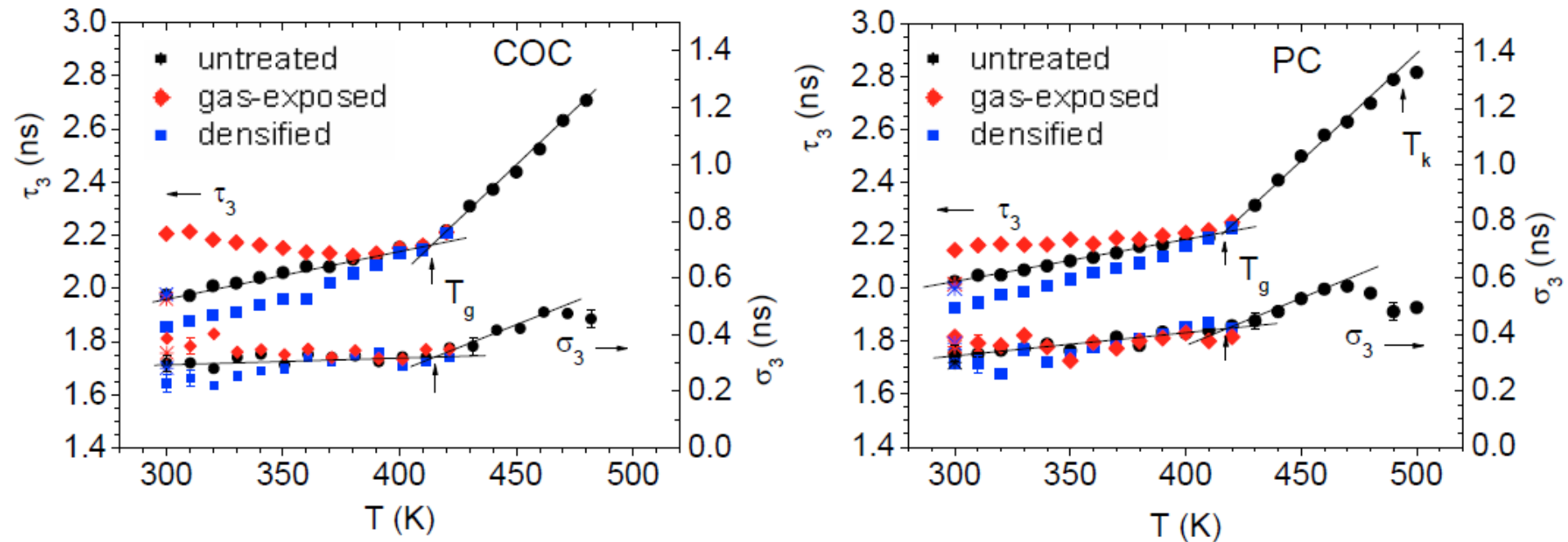


- TE model valid for $r > 2\text{nm}$
- very successful for open-volume characterization in polymers



Tao, S. J. *J. Chem. Phys.* **1972**, *56*, 5499-5510. / Eldrup, M.; Lightbody, D.; Sherwood, J. N. *Chem. Phys.* **1981**, *63*, 51-58.

Polymer research

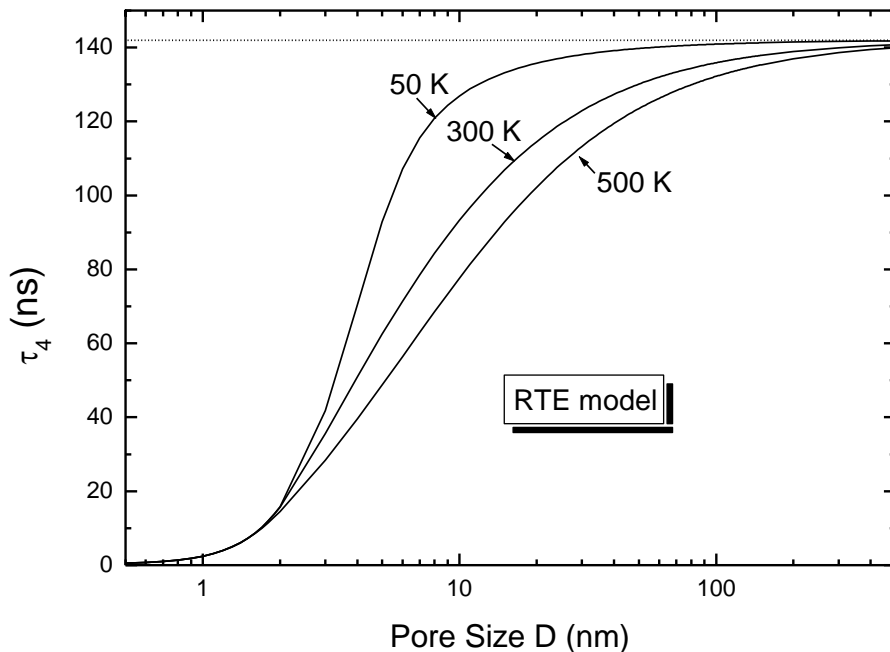


The mean, τ_3 , and the mean dispersion, σ_3 , of *o*-Ps lifetimes as a function of temperature T for densified at 200 MPa (blue), gas-exposed (red) and untreated (black) COC and PC.

Model for $R > 1$ nm - RTE

- Rectangular TE model = RTE model (for 3D cubic pores):

$$\lambda_{RTE}(D, T) = \lambda_A - \frac{\lambda_S - \lambda_{3\gamma}}{4} \left[1 - \frac{2\delta}{D} + \frac{\sum_{i=1}^{\infty} \frac{1}{i\pi} \sin\left(\frac{2i\pi\delta}{D}\right) e^{\left(\frac{-\beta i^2}{D^2 kT}\right)}}{\sum_{i=1}^{\infty} e^{\left(\frac{-\beta i^2}{D^2 kT}\right)}} \right]^3$$



- Boltzmann statistics ascribes explicit temperature dependence to the lifetime
- Rectangular geometry \rightarrow prevention of complicated Bessel functions
- $\delta = 0.18$ nm analogous to TE model

D. W. Gidley, T. L. Dull, W. E. Frieze, J. N. Sun, A. F. Yee, *J. Phys. Chem. B* **2001**, *105*, 4657.

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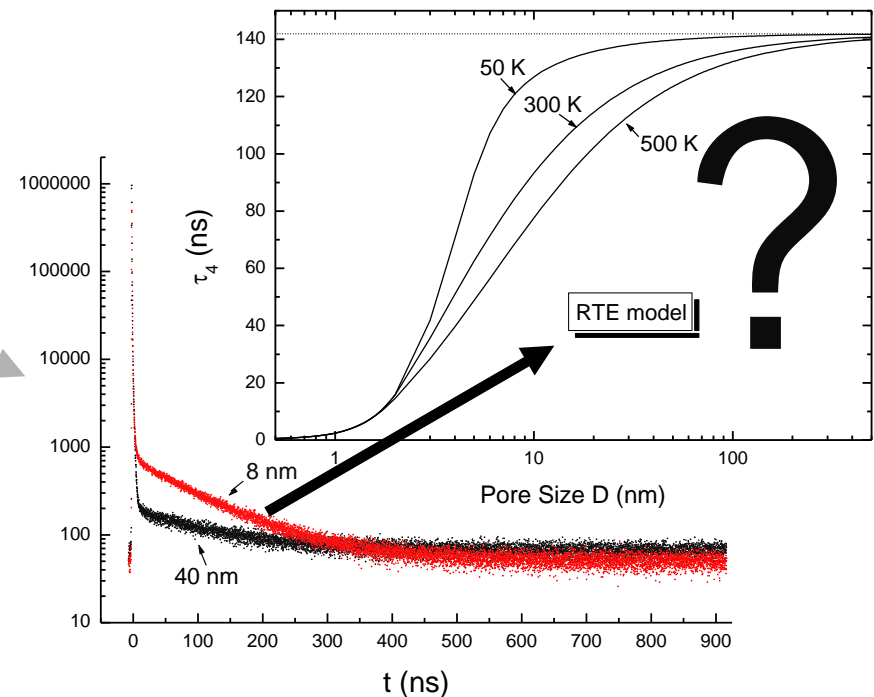
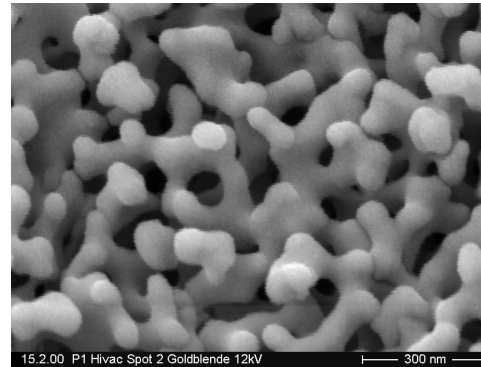


Z

■ Experimental results

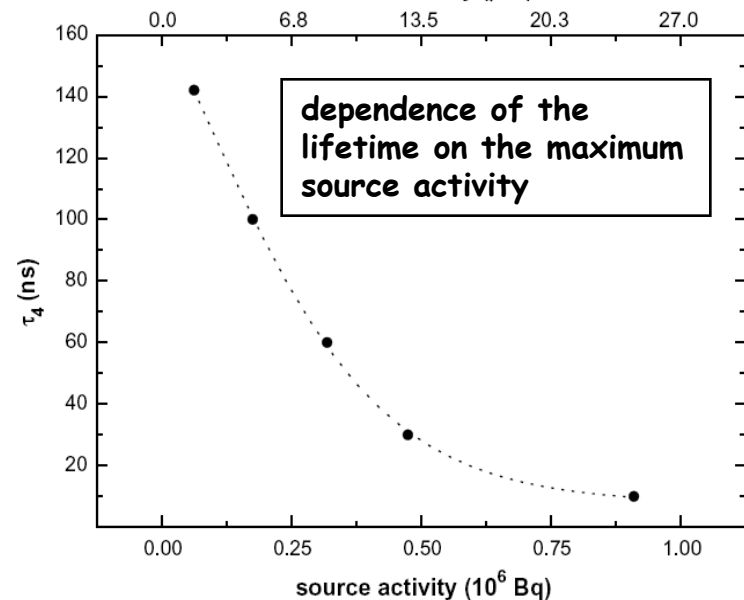
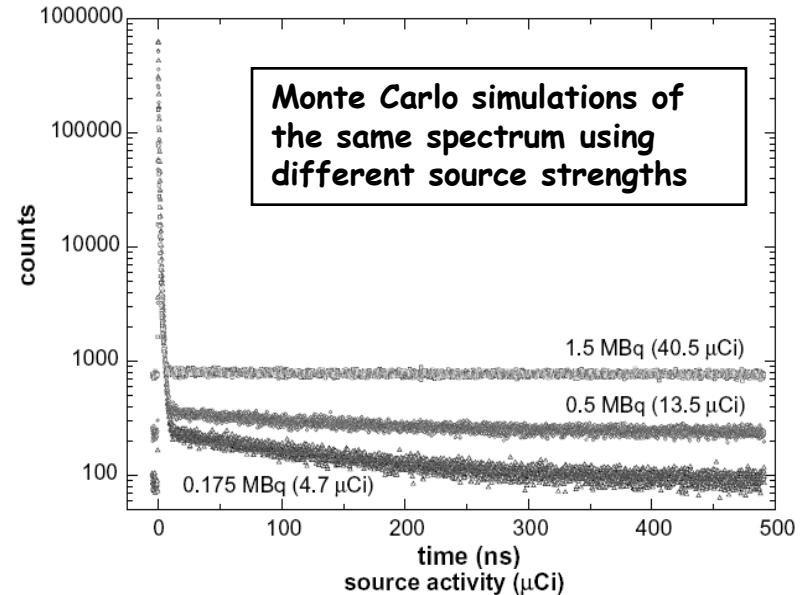
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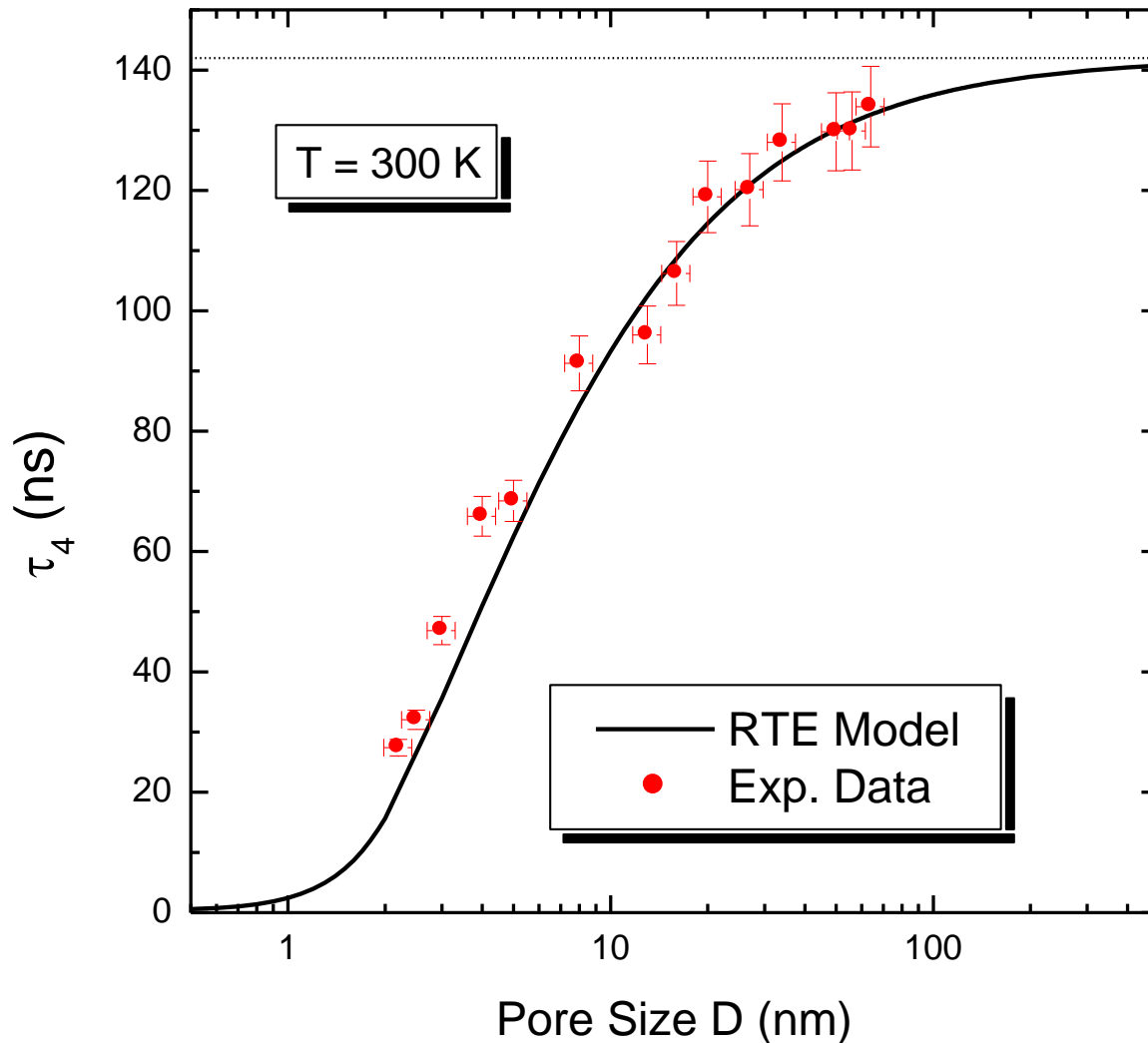


The experiments

- Important: weak source required to obtain o-Positron lifetime properly (long lifetime component disturbed by chance coincidences)
- When expecting a lifetime of e.g. 120 ns \rightarrow max. source strength of 3 μCi recommended
- At first measurements at $T = 300 \text{ K}$ on different pore sizes



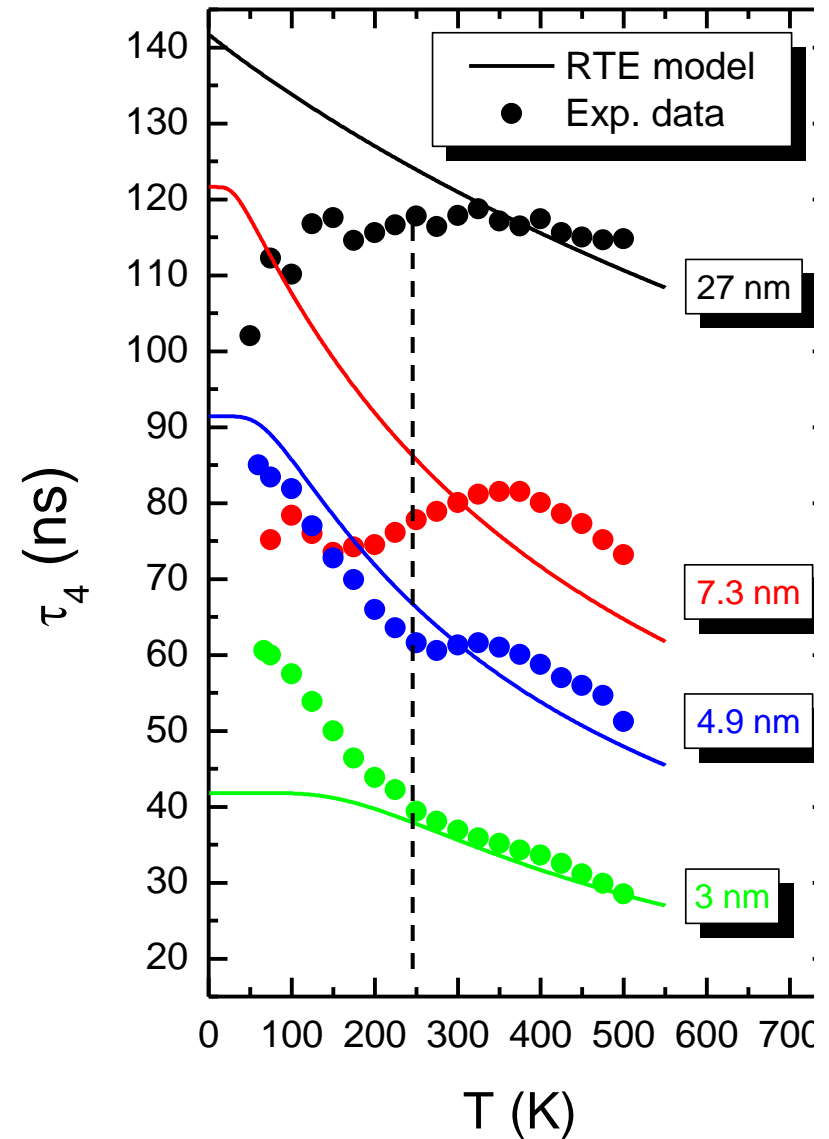
The experiments at $T = 300$ K



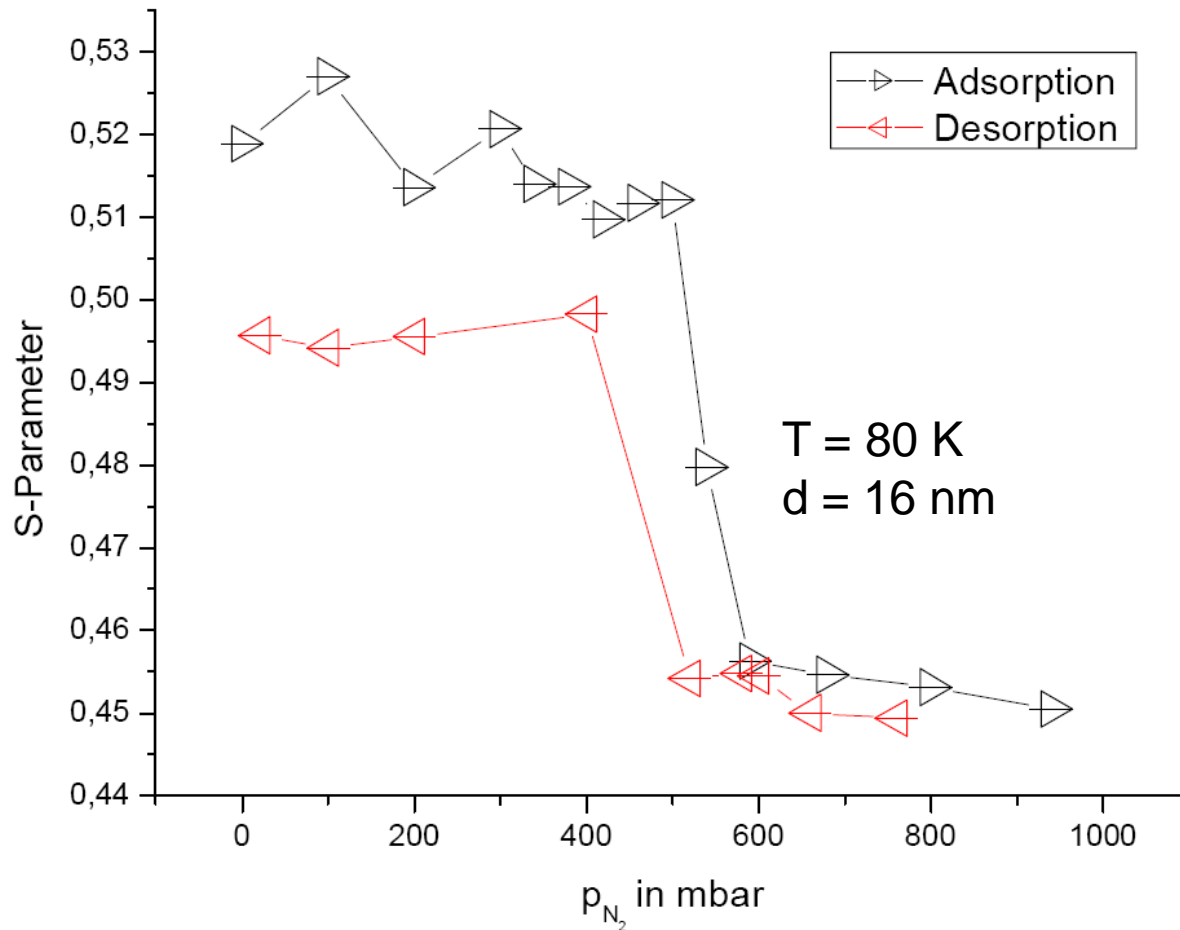
- we measured porous glass in a broad pore size range
- pore size obtained by N_2 -adsorption method
- for $T=300$ K general agreement to the RTE model
- calibration curve for the correlation of o-Ps lifetime and pore size

The T-dependence

- although we found good agreement for $T = 300$ K
- temperature behavior cannot be explained very well at low temperatures
- model too simple



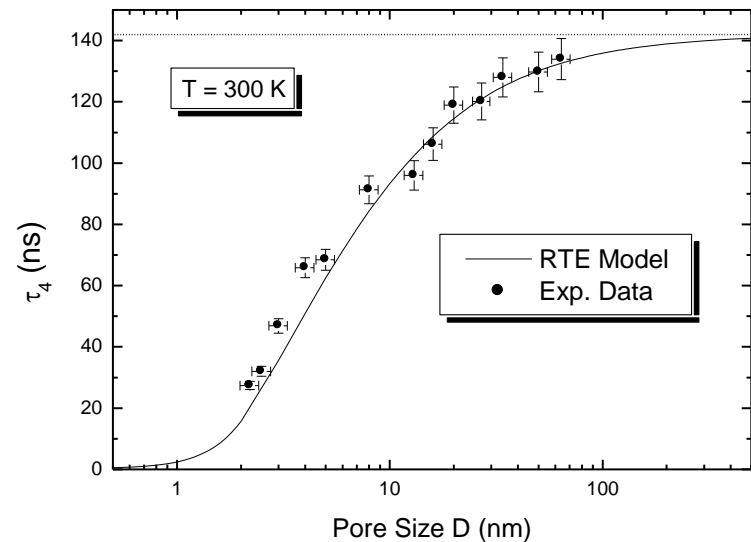
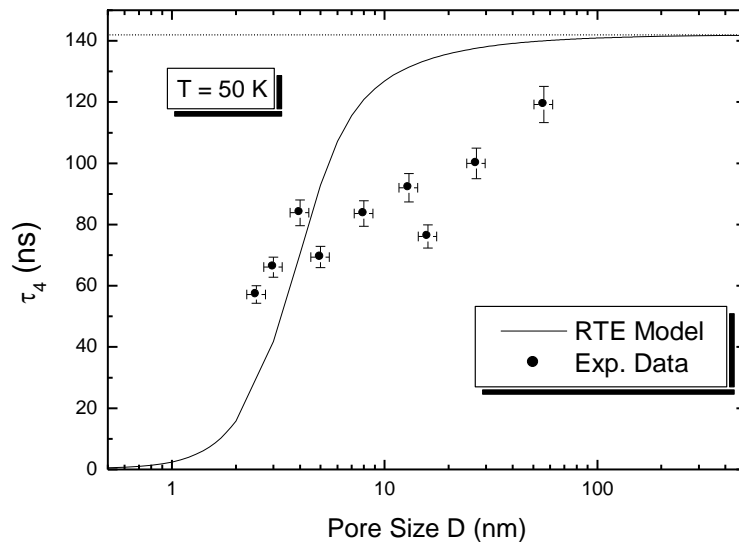
Cryo-condensation in nano-pores



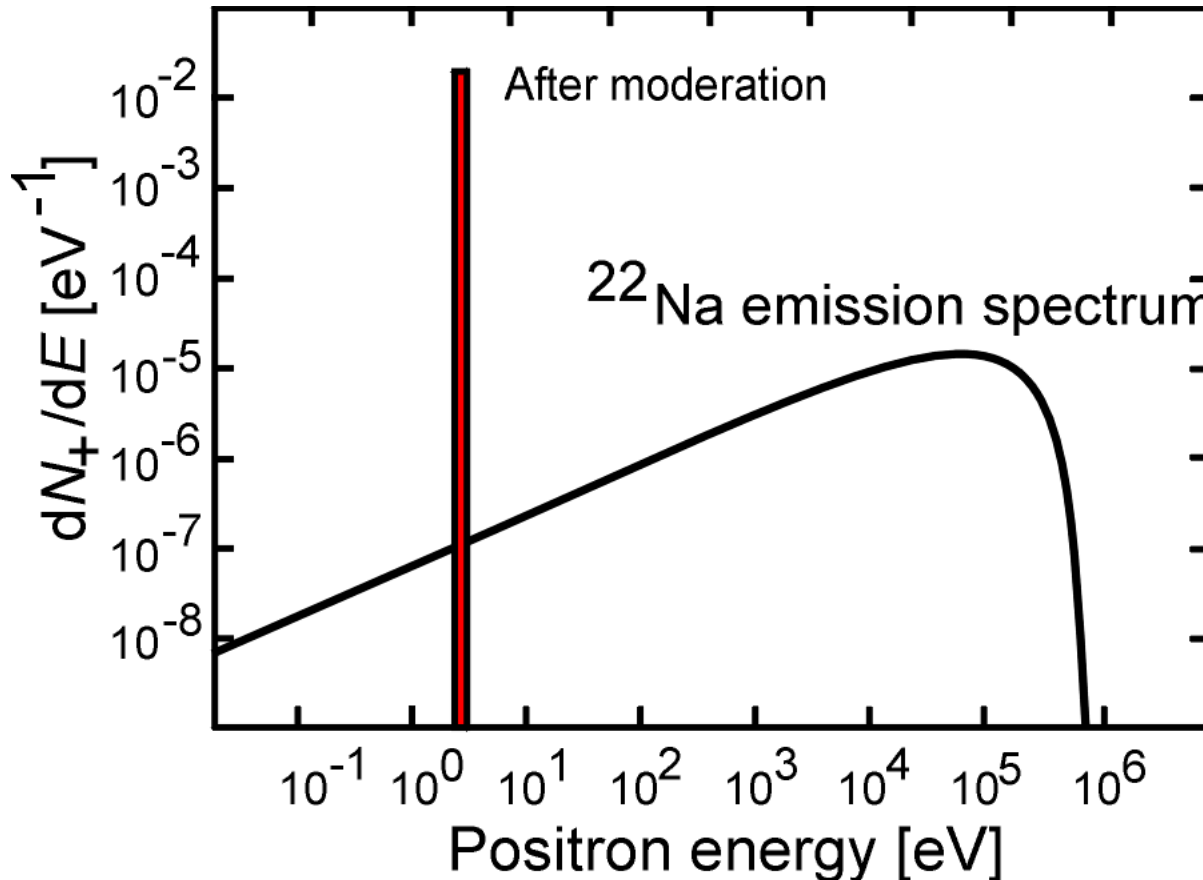
- S-parameter behaves similar like intensity of o- Ps lifetime component
- cryo-condensation can be observed as filling of pores
- phase transition can be studied in a nano-volume as function of size, gas, T and p

Summary

- for $T = 300$ K general agreement to the RTE model \rightarrow at room temperature, PALS is a useful porosimetry tool!
- for $T > 300$ K still acceptable agreement to the RTE model.
- for low temperatures the measurements show disagreement to the RTE model
- Advantages:
 - very sensitive method for small pores (1 nm to 10 nm)
 - also encapsulated pores can be measured
 - non destructive



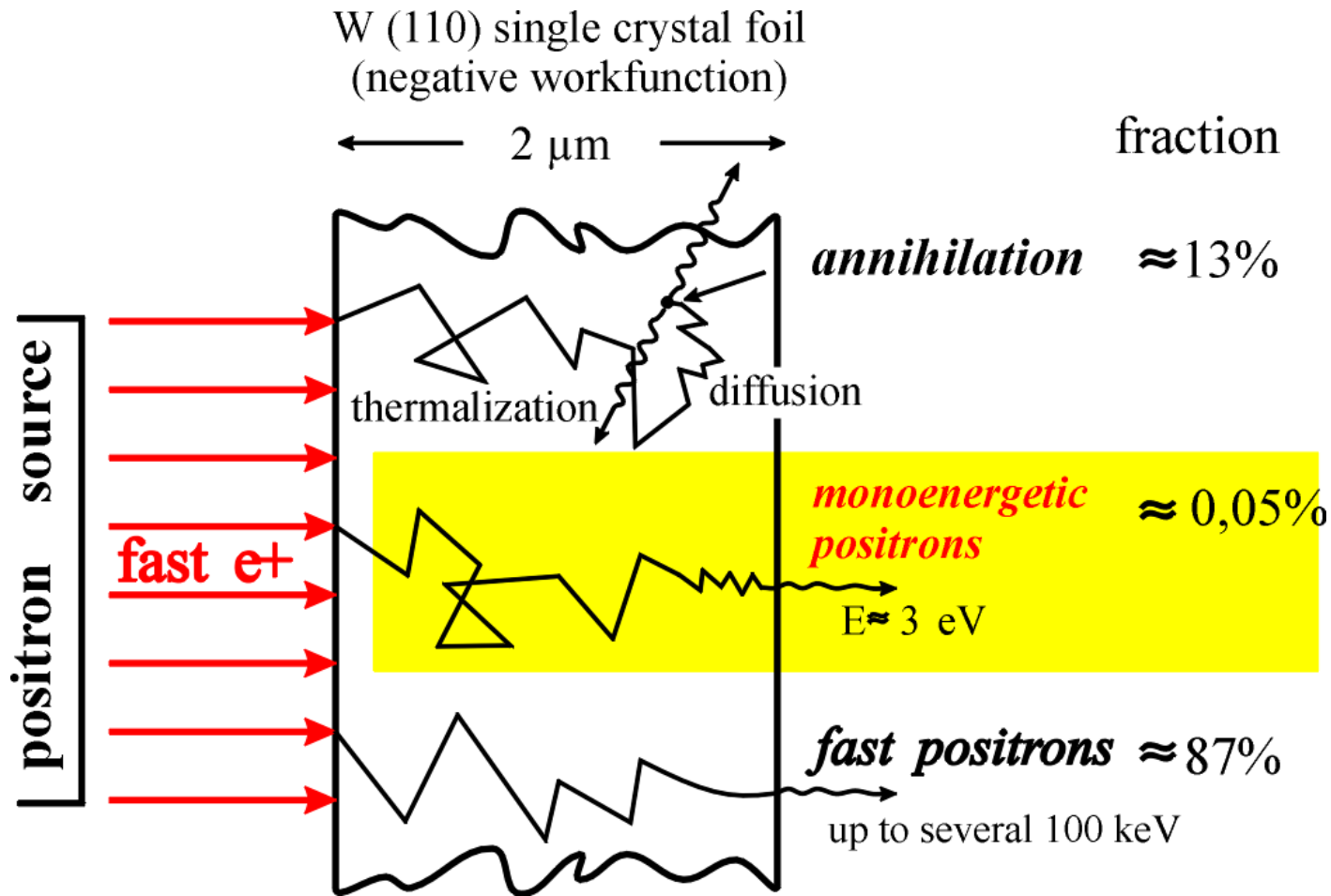
The Slow-Positron Beam Technique



- broad positron emission spectrum from beta sources
- deep implantation into solids
- no use for study of defects in thin layers
- moderation necessary

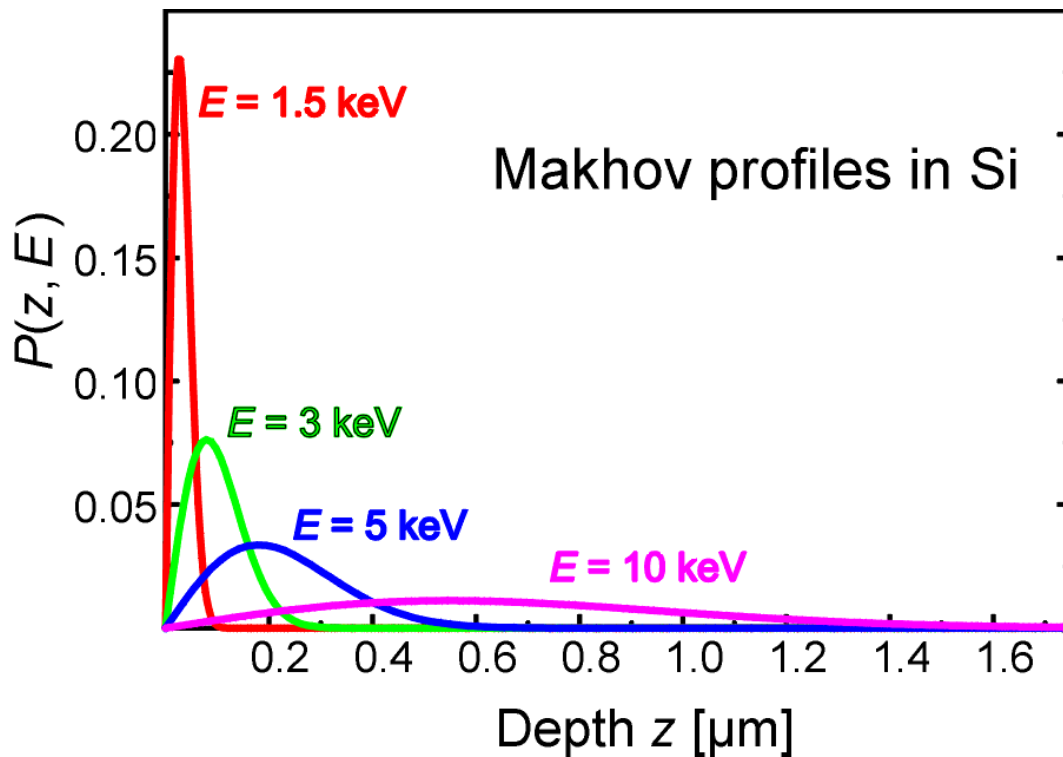
Mean implantation depth of un-moderated positrons ($1/e$): Si: $50\mu\text{m}$

Moderation of Positrons



moderation efficiency: $\approx 10^{-4}$

Implantation Profiles of monoenergetic Positrons



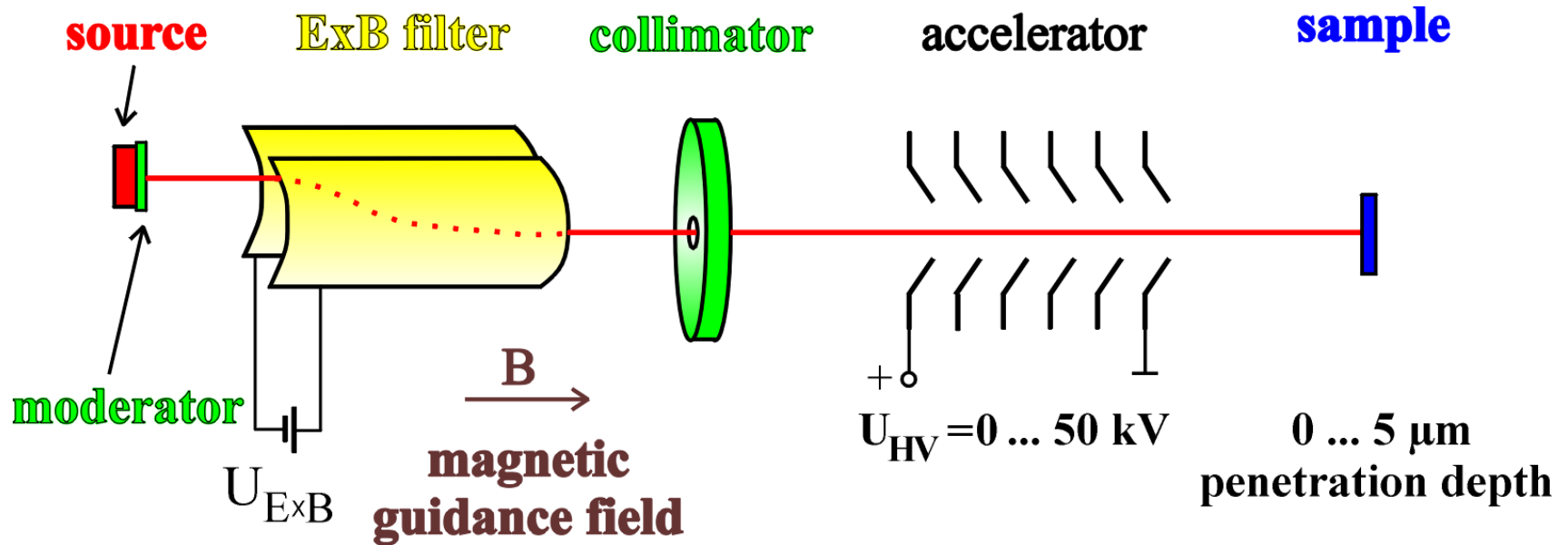
- depth resolution is function of implantation depth
- exact implantation profiles are obtained by Monte-Carlo simulations

$$P(z, E) = \frac{mz^{m-1}}{z_0^m} \exp\left[-\left(\frac{z}{z_0}\right)^m\right]$$

$$z = f(E, \rho) \quad z_0 = \text{const.} \quad m = 2$$

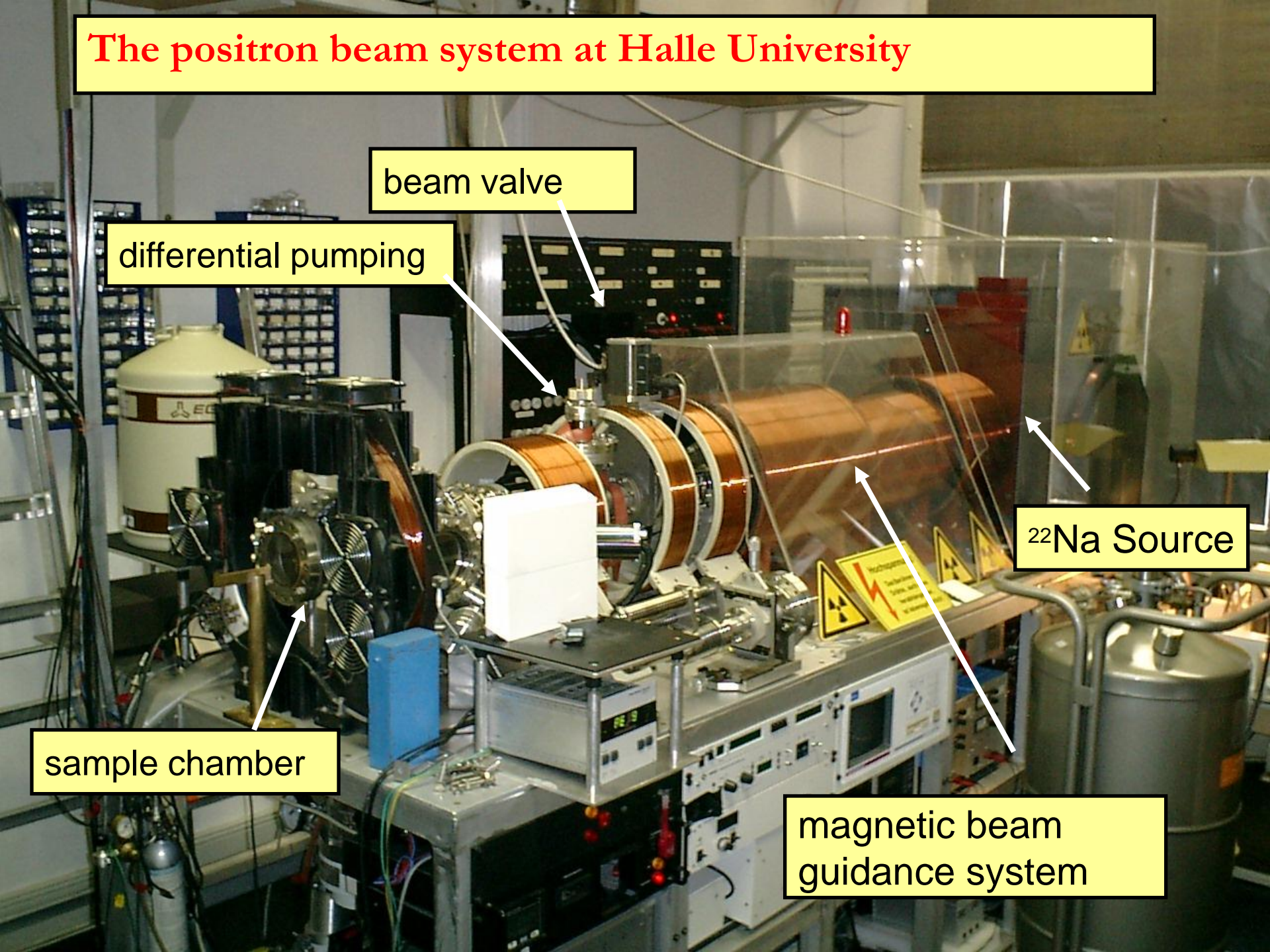
(Makhov, 1961)

The Positron Beam System at Halle University



- spot diameter: 5mm
- time per single Doppler measurement: 20 min
- time per depth scan: 8 hours

The positron beam system at Halle University



beam valve

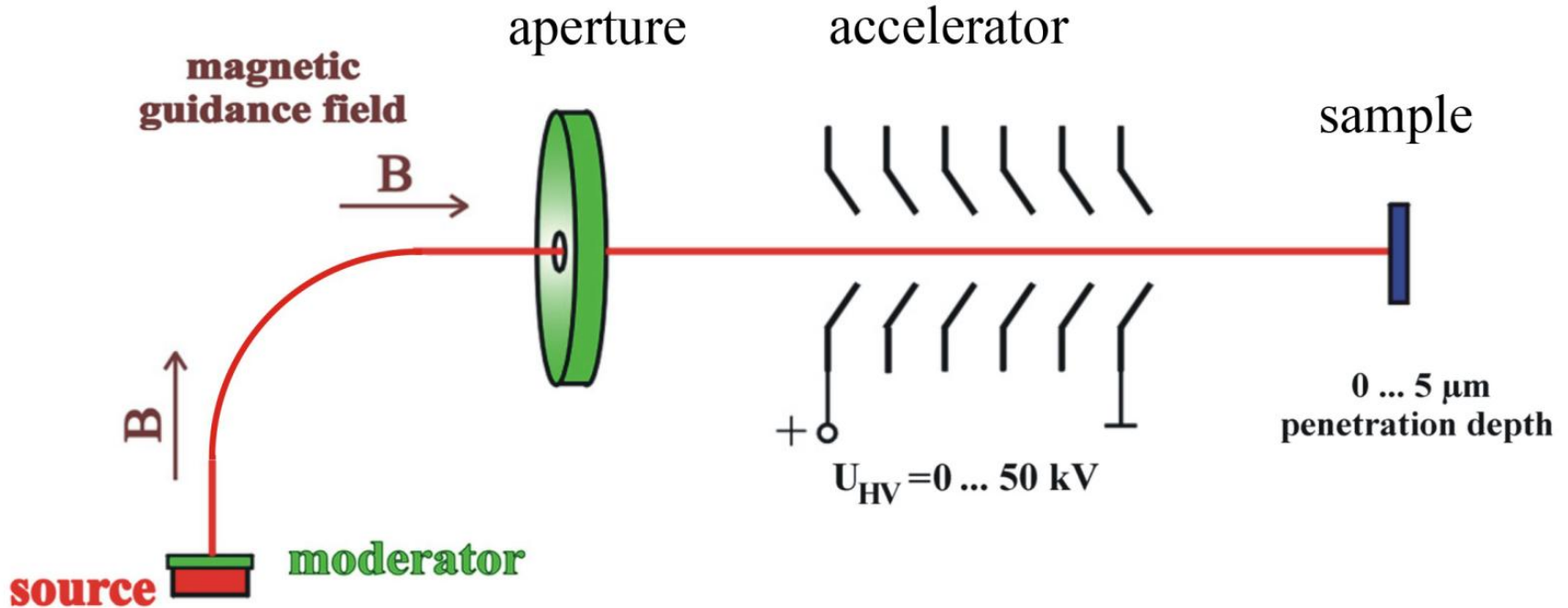
differential pumping

sample chamber

^{22}Na Source

magnetic beam
guidance system

Positron Beam System at Halle University - new setup



- Energy selection now done by a bended tube: slow positrons follow the longitudinal magnetic field
- radiation screening at the floor easier

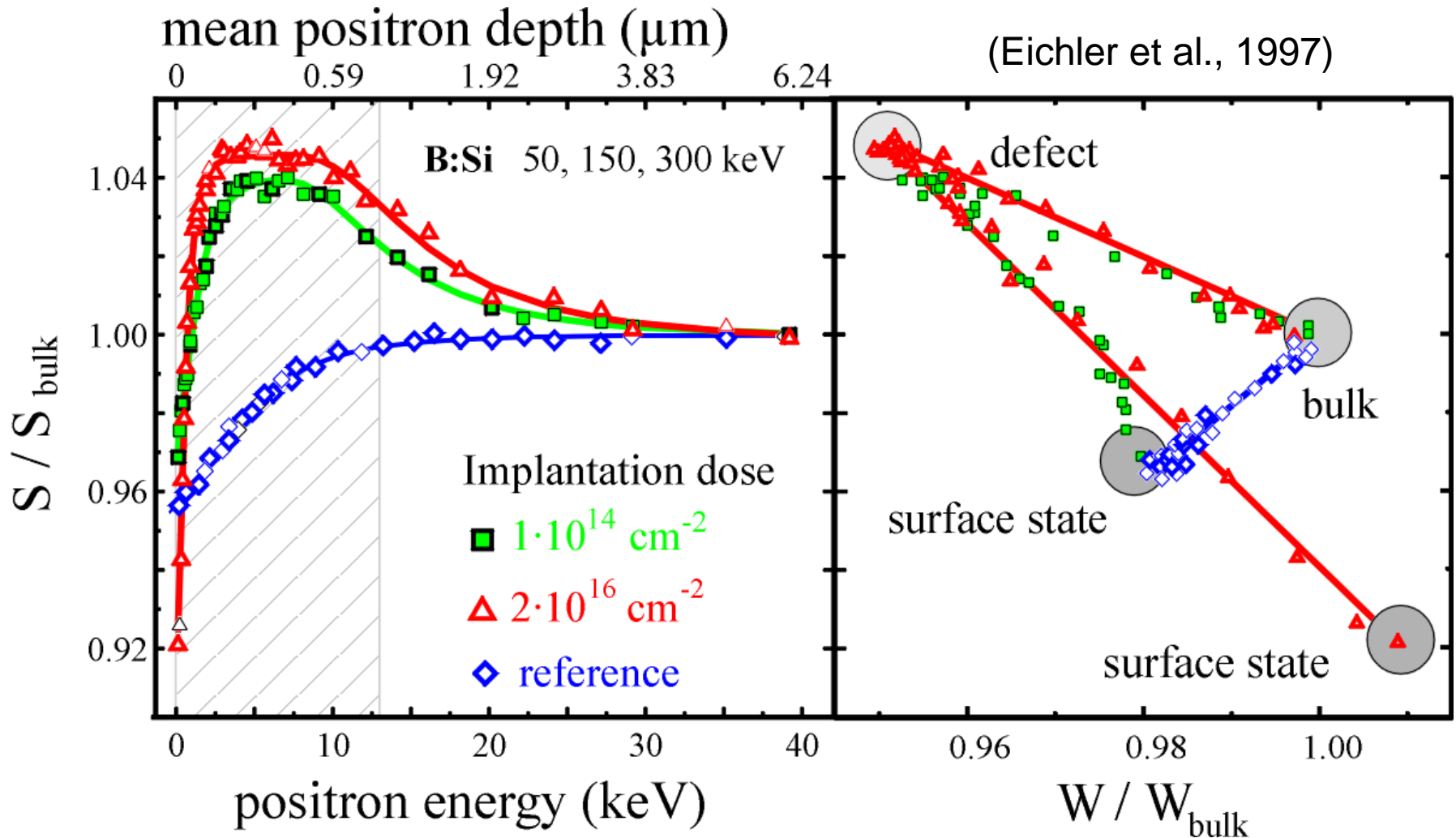
Earth magnetic field must be compensated

- Earth field of $50 \mu\text{T}$ (0.5 G) at a length of 3m and 2 keV beam energy: 25 mm deviation! But most beams are slower...
- Must be compensated
- Compensation necessary only in the two perpendicular directions to the beam
- We used thin Al profile and 10 turns of wire and a few amps



Defects in Si induced by Ion Implantation

- ion implantation is most important doping technique in planar technology
- main problem: generation of defects \Rightarrow positron beam measurements



- Thanks for your patience!

- This talk as pdf?

- <http://positron.physik.uni-halle.de>